# STUDY OF GEOMETRICAL SHAPING OF LINEAR CHAINED POLYMERS STABILIZED AS HELIXES 

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#### Abstract

The hypothesis of the study was that linear chained polymers have a natural chance to stabilize as a helix. Ten linear chained polymers, poly(ethylene glycol), polyethyleneimine, poly(lactic acid), poly(N-vinyl-pyrrolidone), poly(trans-1-butenylene), poly(1-chloro-trans-1-butenylene), poly(1-methyl-trans-1-butenylene), poly(1,4,4-trifluoro-trans-1-butenylene), polyacrylonitrile, polychlorotrifluoroethylene, were investigated. The structure of polymers was drawn and then optimized at Hartree-Fock, 6-31G* level of theory. The helix parameter was extracted from the optimized geometries using a home-made program. Seven out of ten polymers are likely to have a helical structure; the polymers with oxygen shown the highest residual error. The helix coefficient and rotation step per monomer were also calculated. The top three polymers according to rotation step per monomer behaved same as the one according to the helix coefficient. The top three non-increasing order was: polychlorotrifluoroethylene-CI, polyacrylonitrile-N, and poly(lactic acid)-C-methyl. The smallest rotation step per monomer was associated with the smallest value of the helix coefficient (this being linearly related to rotation step per monomer). The highest helix radius was identified for poly(1-chloro-trans-1-butenylene), followed by poly(1-methyl-trans-1-butenylene) and poly(1,4,4-trifluoro-trans-1-butenylene).


Keywords: linear chained polymer, helix radius, computational study

## INTRODUCTION

The polymer term is derived from Greek poly- 'many' \& -mer 'parts' and identifies a molecule constructed by many repeated subunits [1] (Table 1).

[^0]Polymer based materials have different properties, such as resistive memory [2], physical and electronic properties [3], and a variety of uses (see Table $2[4,5]$ ).

Table 1. Basic structure of polymers

|  | Monomer |  | Polymer |  |
| :--- | :--- | :---: | :--- | :--- |
| Structure | Name | CID | Repeating unit | Name |
| $\mathrm{CH}_{2}=\mathrm{CH}_{2}$ | Ethene | $\underline{6325}$ | $(-\mathrm{CH}-\mathrm{CH}-)_{n}$ | Polyethylene |
| $\mathrm{CH}_{2}=\mathrm{CHCl}^{2}$ | Chloroethene | $\underline{6338}$ | $\left(-\mathrm{CH}_{2}-\mathrm{CH}(\mathrm{Cl})-\right)_{n}$ | Polyvinyl chloride |
| $\mathrm{F}_{2} \mathrm{C}=\mathrm{CF}_{2}$ | Tetrafluoroethene | $\underline{8301}$ | $\left(-\mathrm{F}_{2} \mathrm{C}-\mathrm{CF}_{2}-\right)_{n}$ | Polytetrafluoroethylene |

Table 2. Applications of polymers

|  | Polymer | Monomer unit | Uses |
| :---: | :---: | :---: | :---: |
| 1 | polypeptides | -NH-CO- <br> (amino acids) | proteins, wool, silk, steric stabilizers, colloidal additives |
| 2 | polyvinyls | $-\mathrm{CH}_{2}-\mathrm{CH}(\mathrm{X})$ - | plastics |
| 3 | polyesters | -CO-O- | clothing, containers |
| 4 | polysiloxanes, silicones | -Si-O- | lubricants, rubbers, paints |
| 5 | polyamide | -NH-CO- | nylon, fabrics, auto parts |
| 6 | polyurethanes | -NH-CO-O- | adhesives, flexible furniture |
| 7 | cellulose | -C-O- | paper, photographic film |
| 8 | polycarbonates | -O-CO-O- | optical equipment, CDs |
| 9 | polyethylene oxide (PEO, PEG) | - $\mathrm{CH}_{2}-\mathrm{O}-\mathrm{CH}_{2}-$ | detergents, cosmetics |
| 10 | polyacrylamide (PAA) | $-\mathrm{CH}_{2}-\mathrm{CH}\left(\mathrm{CONH}_{2}\right)$ - | plastics, textiles, diapers |
| 11 | polyvinyl alcohol (PVA) | $-\mathrm{CH}_{2}-\mathrm{CH}(\mathrm{OH})$ - | fibers, adhesives, textiles |
| 12 | polyethylene (PE) | $-\mathrm{CH}_{2}-\mathrm{CH}_{2}-$ | coating, containers, films |
| 13 | polystyrene (PS) | $-\mathrm{CH}_{2}-\mathrm{CH}\left(\mathrm{C}_{6} \mathrm{H}_{5}\right)$ - | packing (fast-food containers), housewares |
| 14 | polybutadiene (PBD) | $\begin{aligned} -\mathrm{CH}_{2} & =\mathrm{CH}- \\ \mathrm{CH} & =\mathrm{CH}_{2}- \end{aligned}$ | latex paints, rubbers |
| 15 | polydimethylsiloxane (PDMS) | -Si( $\left.\mathrm{CH}_{3}\right)_{2}$ - $\mathrm{O}-$ | silicone oil, lubricants |
| 16 | polypropylene (PP) | $-\mathrm{CH}_{2}-\mathrm{CH}\left(\mathrm{CH}_{3}\right)$ - | carpets, bottles, wrap films, microwaveable containers |
| 17 | polymethylmethacrylate (PMMA) | $\begin{gathered} -\mathrm{CH}_{2-} \\ \mathrm{CCH}_{3}\left(\mathrm{CO}_{2} \mathrm{CH}_{3}\right)- \end{gathered}$ | transparent windows, plexiglass, perpex |
| 18 | polytetrafluoroethylene (PTFE) | $-\mathrm{CF}_{2}-\mathrm{CF}_{2}{ }^{-}$ | teflon |
| 19 | polyvinyl chloride (PVC) | $-\mathrm{CH}_{2}-\mathrm{CH}(\mathrm{Cl})-$ | plastic sheet, insulation, pipes |

Classification of polymers is done according to vary criteria, such as synthesis method, molecular structure (linear or branched), chemical family (organic or inorganic), etc. [6].

According to their structure, polymers are classified as [7]:

- Linear: a single continuous chain of repeated unit (such as acrylics, nylons, polyethylene, polyvinyl chloride).
- Branched: side chains of repeated unit connecting into the main chain of repeated unit (such as polyethylene). A linear polymer may show branching points and side chains (parts out of the linear chain, like branches off the trunk of a tree).
- Cross-linked: interconnections between chains (such as rubbers)
- Network: many interconnections between chains such that the entire sample is a single molecule (such as epoxies, phenolics)
- Configuration: 3D structure of polymer based on orientations that can be changed just by breaking the bonds

Linear polymers may include different atoms in the main chain and may or may not have rings (cyclolinear polymers):

- Just carbon atoms (C) in the main chain: saturated main chain (polyethylene, polypropylene), unsaturation of the main chain (natural rubber, polyacetylene), or polar side groups (PVC, PMMA, Teflon)
- Oxygen (O) in the main chain: polyformaldehyde (acetal), epoxy polymers (PEO, PPO), polyesters, polycarbonates
- N in the main chain: polyamines, polyamides (polypeptides), polyurethanes
- Other atoms in the main chain: polysulfones, polysiloxanes (silicones), polyphosphazenes
- Rings in the main chain: poly(p-phenylene), polyaniline, cellulose

After a while, linear constructions of polymers (such is a hose) have the natural tendency to bend. The expose of hoses to biodiesel, for example, led to decrease of its tensile strength [8], while the sterilization of surgical silicone hose led to modification of the structure of polymer [9]. The main question is in which degree this tendency to bend is a natural tendency?

A series of studies shown that some synthetic polymers (m-terphenylbased $\pi$-conjugated polymer [10], poly(ethylene glycol) (PEG) [11-13], poly(ethylene imine) (PEI) [14], squaraine polymers [15]) or natural polymerlike compounds (functional polymers like DNA [16], RNA [17], globular proteins [18]) have the tendency to form a helix. Of course, the most important case is the helix of DNA, but the question about the tendency of other polymers to form helix even if was investigated [19-22] it still needs more attention. The present study takes into account a series of polymers, from both biological and synthetic groups having as the subject of investigation the tendency of polymers to form a helix.

## RESULTS AND DISCUSSION

The residual errors obtained by applying (3) varied from 0.0016 to 0.4428 with highest residuals for oxygen atom in any position (even, odd, or all polymer \#4, Table 3). Systematically, the polymers with oxygen had highest values of residuals, followed by the polymers with C-methyl group. The highest value of residual errors is thus observed in the polymer that tend to form a double helix and all investigated structures of \#4 proved to have outliers. Furthermore, the sample size after removal of the outliers (if any) proved inversed linearly related to residual errors (Pearson's correlation coefficient $R=-0.8969, p$-value $=7.68 \mathrm{e}-5$ ). Accordingly, with some exceptions (see \#4 O-all, Table 3), when the sample size increases, the residual error decreases.

Table 3. Residual errors for investigated polymers likely to have a helical structure

| Polymer | Atom | $n$ | n' | Residual error (SS) | Remarks |
| :---: | :---: | :---: | :---: | :---: | :---: |
| \#3 | C-methyl | 17 | 15 | 0.0393 | SS 100 times higher with the ends $\boldsymbol{\rightarrow}$ the ends are outliers (2) |
| \#4 | O-even | 11 | 9 | 0.3123 | SS 46 times higher with the ends $\rightarrow$ the ends are outliers (2) |
|  | O-odd | 12 | 10 | 0.4428 | SS 66 times higher with the ends $\rightarrow$ the ends are outliers (2) |
|  | O-all | 21 | 17 | 0.1255 | SS 100 times higher with the ends $\boldsymbol{\rightarrow}$ the ends are outliers (4) |
| \#6 | Cl | 18 | 18 | 0.0127 | Ends are not outliers |
| \#7 | C-methyl | 18 | 18 | 0.0684 | Ends are not outliers |
| \#8 | C-1 | 18 | 18 | 0.0033 | Ends are not outliers |
|  | C-2 | 18 | 18 | 0.0016 | Ends are not outliers |
|  | C-3 | 18 | 18 | 0.0019 | Ends are not outliers |
|  | C-4 | 18 | 18 | 0.0016 | Ends are not outliers |
| \#9 | N | 18 | 16 | 0.0285 | SS 4 times higher with the ends $\rightarrow$ the ends are outliers (2) |
| \#10 | Cl | 18 | 16 | 0.0047 | SS 37 times higher with the ends $\rightarrow$ the ends these are outliers (2) |

$n$ = sample size; $n^{\prime}=$ sample size after removal of the outliers

The rotation step (coefficient $\mathrm{c}_{0}$ in (2)) and the rotation step/monomer were obtained for each investigated polymer and are given in Table 4. All other calculated coefficients of the helix of the investigated polymers are given in Table 5.

Table 4. The coefficient of the helix and rotation step

| Polymer (atom type) | Helix coefficient (co) | Rotation step (/monomer) $\left(^{\circ}\right.$ ) |
| :---: | :---: | :---: |
| \#3 (- C-methyl) | $2.972 \mathrm{e}-1$ | 107 |
| \#4 (- O-even) | $9.288 \mathrm{e}-2$ | 33.4 |
| \#4 (- O-odd) | $9.193 \mathrm{e}-2$ | 33.1 |
| \#4 (- O-all) | $9.183 \mathrm{e}-2$ | 33.1 |
|  | $1.233 \mathrm{e}-1$ | 44.4 (odd/even) |
| \#6 (- Cl) | $2.148 \mathrm{e}-2$ | 11.1 |
| \#7 (- C-methyl) | $3.085 \mathrm{e}-2$ |  |
| \#8 (- C-1) |  |  |
| \#8 (- C-2) | $2.743 \mathrm{e}-2$ |  |
| \#8 (- C-3) |  |  |
| \#8 (- C-4) | $3.347 \mathrm{e}-1$ | 120.5 |
| \#9 (- N) | $3.609 \mathrm{e}-1$ | 129.9 |
| \#10 (- Cl) |  |  |

Table 5. Helix coefficients - see (2)

| Polymer (atom type) | Helix coefficients for axial projection |  |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  | x axis |  | $y$ axis |  | z axis |
| \#3 | $\mathrm{C}_{1}$ | 3.9983e-2 | $\mathrm{C}_{5}$ | 3.2314 e 0 | $\mathrm{C}_{9}$ | $3.2505 \mathrm{e}+0$ |
|  | $\mathrm{C}_{2}$ | -2.2040e+0 | C6 | -6.7020e-1 | C10 | $9.0419 \mathrm{e}-1$ |
|  | $\mathrm{C}_{3}$ | -1.7105e+0 | $\mathrm{C}_{7}$ | $5.3824 \mathrm{e}-3$ | $\mathrm{C}_{11}$ | -3.0941e-2 |
|  | $\mathrm{C}_{4}$ | $1.5932 \mathrm{e}+1$ | C8 | -1.2359e-1 | $\mathrm{C}_{12}$ | $2.1275 \mathrm{e}-1$ |
| \#4 (even) | $\mathrm{C}_{1}$ | $3.257 \mathrm{e}+0$ | $\mathrm{C}_{5}$ | 2.088 e 0 | $\mathrm{C}_{9}$ | $4.026 \mathrm{e}+0$ |
|  | $\mathrm{C}_{2}$ | -1.807e+0 | $\mathrm{C}_{6}$ | -1.724e0 | $\mathrm{C}_{10}$ | -3.349e+0 |
|  | $\mathrm{C}_{3}$ | $1.345 \mathrm{e}+0$ | $\mathrm{C}_{7}$ | -2.036e0 | $\mathrm{C}_{11}$ | -1.073e-1 |
|  | $\mathrm{C}_{4}$ | $-1.640 \mathrm{e}+1$ | $\mathrm{C}_{8}$ | $2.487 \mathrm{e}+1$ | $\mathrm{C}_{12}$ | $1.270+0$ |
| \#4 (odd) | $\mathrm{C}_{1}$ | $2.584 \mathrm{e}+0$ | C5 | $1.555 \mathrm{e}+0$ | C9 | $2.906 \mathrm{e}+0$ |
|  | $\mathrm{C}_{2}$ | $-3.290 \mathrm{e}+0$ | $\mathrm{C}_{6}$ | $-3.228 \mathrm{e}+0$ | $\mathrm{C}_{10}$ | -4.865e+0 |
|  | $\mathrm{C}_{3}$ | $1.306 \mathrm{e}+0$ | $\mathrm{C}_{7}$ | -2.039e+0 | $\mathrm{C}_{11}$ | $-9.570 \mathrm{e}-2$ |
|  | $\mathrm{C}_{4}$ | $-1.525 \mathrm{e}+1$ | $\mathrm{C}_{8}$ | $2.358 \mathrm{e}+1$ | $\mathrm{C}_{12}$ | $1.162 \mathrm{e}+0$ |
| \#4 (all) | $\mathrm{C}_{1}$ | $2.9031 \mathrm{e}+0$ | C5 | $1.8583 \mathrm{e}+0$ | C9 | $3.4290 \mathrm{e}+0$ |
|  | $\mathrm{C}_{2}$ | $-2.5169 \mathrm{e}+0$ | $\mathrm{C}_{6}$ | -2.4382e+0 | $\mathrm{C}_{10}$ | $-4.0614 \mathrm{e}+0$ |
|  | $\mathrm{C}_{3}$ | $1.3204 \mathrm{e}+0$ | $\mathrm{C}_{7}$ | -2.0466e+0 | $\mathrm{C}_{11}$ | -1.0094e-1 |
|  | $\mathrm{C}_{4}$ | -1.5757e+1 | C8 | $2.4304 \mathrm{e}+1$ | $\mathrm{C}_{12}$ | $1.2071 \mathrm{e}+0$ |
|  | $\mathrm{C}_{13}$ | $4.1484 \mathrm{e}-1$ | $\mathrm{C}_{14}$ | $2.5537 \mathrm{e}-1$ | $\mathrm{C}_{15}$ | $5.3846 \mathrm{e}-1$ |
|  | $\mathrm{C}_{16}$ | -4.0826e-1 | $\mathrm{C}_{17}$ | $6.6823 \mathrm{e}-1$ | $\mathrm{C}_{18}$ | $2.4509 \mathrm{e}-2$ |
| \# (6) | $\mathrm{C}_{1}$ | 2.48427 e 1 | $\mathrm{C}_{5}$ | $3.19611 \mathrm{e}+1$ | $\mathrm{C}_{9}$ | $1.79865 \mathrm{e}+1$ |
|  | $\mathrm{C}_{2}$ | -2.88747e0 | C6 | -1.34873e+0 | $\mathrm{C}_{10}$ | $1.64965 \mathrm{e}-1$ |
|  | $\mathrm{C}_{3}$ | 1.74459 e 0 | $\mathrm{C}_{7}$ | -1.01784e-1 | $\mathrm{C}_{11}$ | $1.90010 \mathrm{e}+0$ |
|  | $\mathrm{C}_{4}$ | -1.68632e1 | $\mathrm{C}_{8}$ | -2.24718e+1 | $\mathrm{C}_{12}$ | -1.90634e+1 |
| \# (7) | $\mathrm{C}_{1}$ | 1.82640 e 1 | $\mathrm{C}_{5}$ | $2.33646 \mathrm{e}+1$ | C9 | $1.41794 \mathrm{e}+1$ |
|  | $\mathrm{C}_{2}$ | 2.93375 e 0 | $\mathrm{C}_{6}$ | $1.33951 \mathrm{e}+0$ | $\mathrm{C}_{10}$ | -1.99841e-1 |
|  | $\mathrm{C}_{3}$ | 1.57034 e 0 | $\mathrm{C}_{7}$ | $6.32800 \mathrm{e}-2$ | $\mathrm{C}_{11}$ | $1.93730 \mathrm{e}+0$ |
|  | $\mathrm{C}_{4}$ | -1.46022e1 | $\mathrm{C}_{8}$ | $1.16876+1$ | $\mathrm{C}_{12}$ | $-1.77684 \mathrm{e}+1$ |

Table 5. (Continued)

| Polymer <br> (atom type) | Helix coefficients for axial projection |  |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | $\mathrm{C}_{1}$ | $-9.14110 \mathrm{e}+0$ | $\mathrm{C}_{5}$ | $1.51182 \mathrm{e}+1$ | $\mathrm{C}_{9}$ | $1.70206 \mathrm{e}+1$ |
|  | $\mathrm{C}_{2}$ | $-1.46653 \mathrm{e}-1$ | $\mathrm{C}_{6}$ | $-6.30651 \mathrm{e}-1$ | $\mathrm{C}_{10}$ | $1.07360 \mathrm{e}+0$ |
|  | $\mathrm{C}_{3}$ | $3.29307 \mathrm{e}+0$ | $\mathrm{C}_{7}$ | $1.82544 \mathrm{e}+0$ | $\mathrm{C}_{11}$ | $7.95252 \mathrm{e}-1$ |
|  | $\mathrm{C}_{4}$ | $-3.25477 \mathrm{e}+1$ | $\mathrm{C}_{8}$ | $-2.22678 \mathrm{e}+1$ | $\mathrm{C}_{12}$ | $2.49521 \mathrm{e}+0$ |
|  | $\mathrm{C}_{1}$ | $-9.46739 \mathrm{e}+0$ | $\mathrm{C}_{5}$ | $1.57384 \mathrm{e}+1$ | $\mathrm{C}_{9}$ | $1.76728 \mathrm{e}+1$ |
|  | $\mathrm{C}_{2}$ | $-9.44889 \mathrm{e}-2$ | $\mathrm{C}_{6}$ | $-5.76774 \mathrm{e}-1$ | $\mathrm{C}_{10}$ | $1.12318 \mathrm{e}+0$ |
|  | $\mathrm{C}_{3}$ | $3.29773 \mathrm{e}+0$ | $\mathrm{C}_{7}$ | $1.83040 \mathrm{e}+0$ | $\mathrm{C}_{11}$ | $8.01570 \mathrm{e}-1$ |
|  | $\mathrm{C}_{4}$ | $-3.20550 \mathrm{e}+1$ | $\mathrm{C}_{8}$ | $-2.20244 \mathrm{e}+1$ | $\mathrm{C}_{12}$ | $2.53498 \mathrm{e}+0$ |
|  | $\mathrm{C}_{1}$ | $-9.40050 \mathrm{e}+0$ | $\mathrm{C}_{5}$ | $1.56074 \mathrm{e}+1$ | $\mathrm{C}_{9}$ | $1.75550 \mathrm{e}+1$ |
| \#8 (-C3) | $\mathrm{C}_{2}$ | $-7.59906 \mathrm{e}-2$ | $\mathrm{C}_{6}$ | $-5.58858 \mathrm{e}-1$ | $\mathrm{C}_{10}$ | $1.13939 \mathrm{e}+0$ |
|  | $\mathrm{C}_{3}$ | $3.29710 \mathrm{e}+0$ | $\mathrm{C}_{7}$ | $1.82896 \mathrm{e}+0$ | $\mathrm{C}_{11}$ | $8.05768 \mathrm{e}-1$ |
|  | $\mathrm{C}_{4}$ | $-3.07980 \mathrm{e}+1$ | $\mathrm{C}_{8}$ | $-2.12904 \mathrm{e}+1$ | $\mathrm{C}_{12}$ | $2.81228 \mathrm{e}+0$ |
|  | $\mathrm{C}_{1}$ | $-9.18430 \mathrm{e}+0$ | $\mathrm{C}_{5}$ | $1.53022 \mathrm{e}+1$ | $\mathrm{C}_{9}$ | $1.71931 \mathrm{e}+1$ |
| \#8 (-C4) | $\mathrm{C}_{2}$ | $6.26188 \mathrm{e}-3$ | $\mathrm{C}_{6}$ | $-4.76742 \mathrm{e}-1$ | $\mathrm{C}_{10}$ | $1.22481 \mathrm{e}+0$ |
|  | $\mathrm{C}_{3}$ | $3.30040 \mathrm{e}+0$ | $\mathrm{C}_{7}$ | $1.82834 \mathrm{e}+0$ | $\mathrm{C}_{11}$ | $8.03362 \mathrm{e}-1$ |
|  | $\mathrm{C}_{4}$ | $-3.06020 \mathrm{e}+1$ | $\mathrm{C}_{8}$ | $-2.12218 \mathrm{e}+1$ | $\mathrm{C}_{12}$ | $2.87783 \mathrm{e}+0$ |
|  | $\mathrm{C}_{1}$ | $-3.21292 \mathrm{e}+0$ | $\mathrm{C}_{5}$ | $1.08286 \mathrm{e}-2$ | $\mathrm{C}_{9}$ | $3.22894 \mathrm{e}+0$ |
| \#9 (-N) | $\mathrm{C}_{2}$ | $-2.35726 \mathrm{e}+0$ | $\mathrm{C}_{6}$ | $-2.48275 \mathrm{e}+0$ | $\mathrm{C}_{10}$ | $2.35661 \mathrm{e}+0$ |
|  | $\mathrm{C}_{3}$ | $3.07394 \mathrm{e}-3$ | $\mathrm{C}_{7}$ | $2.21476 \mathrm{e}+0$ | $\mathrm{C}_{11}$ | $7.50670 \mathrm{e}-4$ |
|  | $\mathrm{C}_{4}$ | $-2.71106 \mathrm{e}-2$ | $\mathrm{C}_{8}$ | $-1.96284 \mathrm{e}+1$ | $\mathrm{C}_{12}$ | $1.54767 \mathrm{e}-3$ |
|  | $\mathrm{C}_{1}$ | $2.42189 \mathrm{e}+0$ | $\mathrm{C}_{5}$ | $-3.60496 \mathrm{e}-2$ | $\mathrm{C}_{9}$ | $-2.41754 \mathrm{e}+0$ |
| \#10 (-Cl) | $\mathrm{C}_{2}$ | $-7.72781 \mathrm{e}-1$ | $\mathrm{C}_{6}$ | $-7.46450 \mathrm{e}-1$ | $\mathrm{C}_{10}$ | $-2.34239 \mathrm{e}+0$ |
|  | $\mathrm{C}_{3}$ | $2.90532 \mathrm{e}-2$ | $\mathrm{C}_{7}$ | $2.32023 \mathrm{e}+0$ | $\mathrm{C}_{11}$ | $3.25795 \mathrm{e}-3$ |
|  | $\mathrm{C}_{4}$ | $-2.59285 \mathrm{e}-1$ | $\mathrm{C}_{8}$ | $-2.07402 \mathrm{e}+1$ | $\mathrm{C}_{12}$ | $1.17230 \mathrm{e}-2$ |

The smallest coefficient of the helix was observed on polymers with highest size ( $\mathrm{n}=18$, see Table 4); and without outliers while the highest coefficient of the helix was observed in polymers \#3, \#9 and \#10. As expected, the smallest rotation step per monomer was associated with the smallest value of the helix coefficient, which is linearly related to the rotation step per monomer ( $R=0.9999$, $p$-value $=8.55 \mathrm{e}-25$ ).

Several coefficients, such as $\mathrm{c}_{4}, \mathrm{c}_{8}, \mathrm{c}_{12}$ (and $\mathrm{c}_{16}, \mathrm{c}_{17}, \mathrm{c}_{18}$ for the double helix) are not of interest since represent the position of the helix relative to the origin of the coordinates system, position which is arbitrary. The rest of coefficients represent the projection of the helix on a coordinate system rotated relatively to the convenient direction of the helix propagation.

Three coefficients (among those provided in Table 5) are of interest: $\mathrm{c}_{1}, \mathrm{c}_{5}$ and $\mathrm{c}_{9}$ since they allow computing the approximation of the helix radius, by formula:

$$
\begin{equation*}
\text { helix radius }=\frac{\sqrt{c_{1}^{2}+c_{5}^{2}+c_{9}^{2}}}{\sqrt{2}} \tag{5}
\end{equation*}
$$

The smallest helix radius was evaluated for \#10 (internal helix), followed by \#4odd (double helix), and \#3 (internal helix) (see Table 6). The highest value was estimated for the polymer \#6, followed by \#7 and \#8, all polymers being with sample size of 18 and without any outlier.

Starting from the values obtained in this research, the coefficients of the 'classical' helix equation (0) could be obtained. Furthermore, the results obtained here could be used to investigate other polymers likely to form helixes.

## CONCLUSIONS

Seven out of ten investigated polymers are likely to form helixes (internal and double helix). The polymer with the highest coefficient of the helix and highest rotation step per monomer was a polymer with internal helix, polychlorotrifluoroethylene-Cl. The coefficient of the helix proved to be linearly related to the rotation step per monomer ( $p$-value $<0.0001$ ).

Table 6. Radius of the helix approximated by (5)

| Polymer (atom type) | Radius (Angs) |
| :---: | :---: |
| $\# 3$ | 3.24 |
| $\# 4($ even $)$ | 3.95 |
| $\# 4$ (odd) | 2.96 |
| $\# 4$ (all) | 3.44 |
| $\# 6$ | 31.32 |
| $\# 7$ | 23.24 |
| $\# 8(-\mathrm{C} 1)$ | 17.35 |
| $\# 8(-\mathrm{C} 2)$ | 18.02 |
| $\# 8(-\mathrm{Cl})$ | 17.89 |
| $\# 8(-\mathrm{C} 4)$ | 17.52 |
| $\# 9(-\mathrm{N})$ | 3.22 |
| $\# 10(-\mathrm{Cl})$ | 2.42 |

The highest helix radius was identified for poly(1-chloro-trans-1-butenylene) while the smallest helix radius was calculated for polychlorotrifluoroethylene-CI.

## EXPERIMENTAL SECTION

Hypothesis: Linear chained polymers have a good chance to stabilize as a helix. A series of linear polymers were included into this study. Not all of them have enough steric constraints to provide a 'regular irregularity', the essential ingredient of the helix form. Table 7 gives the series of investigated polymers, along with this first level of approximation revealed for instance by a simple geometry built at any level of theory.

Table 7. Structural characteristics and use of the investigated linear polymers

|  | Name | Structure | Remarks | Likely to provide helix |
| :---: | :---: | :---: | :---: | :---: |
| 1 | poly(ethylene glycol) |  | PEG - biological interest (laxative [23], improvement of action potential after spinal cord injury [24], drug delivery [25]) | [ NO ] purely linear |
| 2 | polyethyleneimine |  | PEI - biological interest (gene delivery [26], drug carrier [27]) | [ NO ] <br> purely linear |
| 3 | poly(lactic acid) |  | PLA - biological interest [28] (tissue engineering [29], suture materials [30], delivery systems [31]) | [YES], <br> internal helix <br> (see Fig. 2) |
| 4 | poly( N -vinyl-pyrrolidone) |  | PVP - other interest (beverages, disinfectant, in complexation with of iodine, film forming agents $[32,33]$ ) | [YES] <br> double helix (see Fig. 3) |
|  | poly(trans-1-butenylene) |  | other interest (glass transition temperature [34]) | [YES] |
|  | poly(1-chloro-trans-1butenylene) |  | other interest (glass transition temperature [34]) | [YES] |
| 7 | poly(1-methyl-trans-1butenylene) |  | other interest (glass transition temperature [34]) | [YES] |
|  | poly(1,4,4-trifluoro-trans-1-butenylene) |  | other interest (glass transition temperature [34]) | [YES] |
| 9 | polyacrylonitrile |  | PAN - other interest (nanofibers [35,36], treatment of metals [36]) | [YES] <br> internal helix <br> (see Fig. 4) |
| 10 | Polychlorotrifluoroethylene |  | PCTFE or PTFCE other interest (chemical industry, manufacturing, electronics, architecture, energy, health and domestic sectors [37]) | [YES] <br> internal helix <br> (see Fig. 5) |

Spartan optimized geometries are shown in Figure 1.




Polymer
\#8

Polymer \#9

Polymer \#10


Figure 1. Spartan images of the studied polymers with optimized geometry


Figure 2. Internal helix of poly(lactic acid) (O atoms, red)


Figure 3. Double helix of poly( N -vinyl-pyrrolidone) ( N atoms, blue; O atoms, red)


Figure 4. Internal helix of polyacrylonitrile ( N atoms, blue)


Figure 5. Internal helix of polychlorotrifluoroethylene (chlorine atoms, green, fluorine atoms, cyan)

The 10 polymers of interest were drawn by HyperChem and the geometry of each polymer was optimized. The optimization was done by Spartan software, at Hartree-Fock (HF [38]) level of theory ( $6-31 \mathrm{G}^{*}$ [39]). A home-made program, that extracts from the optimized geometry the parameters of the helix.

Recall that a helix is defined as in Eq 0 :

$$
\begin{align*}
& x=a_{1}+a_{2} \cdot \sin \left(a_{0} \cdot t\right) \\
& y=a_{3}+a_{4} \cdot \cos \left(a_{0} \cdot t\right)  \tag{0}\\
& z=a_{5}+a_{6} \cdot t
\end{align*}
$$

where $x, y$, and $z$ are the axial projections of the helix, and $a_{i}(0 \leq i \leq 6)$ are unknown coefficients ( $a_{1}, a_{3}$ and $a_{5}$ expressing a translation of the reference system relative to the helix, by keeping the direction of the axis the same; $a_{0}$, $\mathrm{a}_{2}, \mathrm{a}_{4}$ and $\mathrm{a}_{6}$ are helix parameters; for circular-based helixes $\mathrm{a}_{2}=\mathrm{a}_{4}$ ); t defines the evolution of the helix.

Equation 0 is theoretically important, but in practice there is no way to align the helix on the $z$-axis without the identification of the unknown coefficients ( $\mathrm{a}_{0} . . \mathrm{a}_{6}$ ); actually the equations of the helix are hidden behind a transformation of the system coordinates. Therefore, the equations maximizing the agreement between the model and the observation may be as follows:

$$
\begin{align*}
& x=b_{1} \cdot \sin \left(b_{0} \cdot t\right)+b_{2} \cdot \cos \left(b_{0} \cdot t\right)+b_{3} \cdot t+b_{4} \\
& y=b_{5} \cdot \sin \left(b_{0} \cdot t\right)+b_{6} \cdot \cos \left(b_{0} \cdot t\right)+b_{7} \cdot t+b_{8}  \tag{1}\\
& z=b_{9} \cdot \sin \left(b_{0} \cdot t\right)+b_{10} \cdot \cos \left(b_{0} \cdot t\right)+b_{11} \cdot t+b_{12}
\end{align*}
$$

where $b_{i}(0 \leq i \leq 12)$ are unknown coefficients (to be determined), and $t$ is as above.

Some unknown coefficients may be incorporated inside of the periodic functions, providing a more convenient expression for the helix:

$$
\begin{align*}
& x=c_{1} \cdot \cos \left(c_{0} \cdot 2 \pi \cdot t+2 \pi \cdot c_{2}\right)+c_{3} \cdot t+c_{4} \\
& y=c_{5} \cdot \cos \left(c_{0} \cdot 2 \pi \cdot t+2 \pi \cdot c_{6}\right)+c_{7} \cdot t+c_{8}  \tag{2}\\
& z=c_{9} \cdot \cos \left(c_{0} \cdot 2 \pi \cdot t+2 \pi \cdot c_{10}\right)+c_{11} \cdot t+c_{12}
\end{align*}
$$

where again $\mathrm{c}_{\mathrm{i}}(0 \leq \mathrm{i} \leq 12)$ are unknown coefficients, and $2 \pi$ coefficient is introduced for convenient interpretation of the $c_{0}$ and $c_{2}, c_{6}$ and $c_{10}$ coefficients.

The maximization of the agreement was achieved by minimizing the squared residual sum:

$$
\begin{align*}
& \mathrm{S}^{2}=\mathrm{S}_{\mathrm{x}, \mathrm{i}}^{2}+\mathrm{S}_{\mathrm{y}, \mathrm{i}}^{2}+\mathrm{S}_{\mathrm{y}, \mathrm{i}}^{2} \rightarrow \min \\
& \mathrm{~S}_{\mathrm{x}, \mathrm{i}}^{2}=\sum_{\mathrm{i}=1}^{\mathrm{n}}\left(\mathrm{x}_{\mathrm{i}}-\mathrm{c}_{1} \cdot \cos \left(\mathrm{c}_{0} \cdot \mathrm{t}+\mathrm{c}_{2}\right)-\mathrm{c}_{3} \cdot \mathrm{t}-\mathrm{c}_{4}\right)^{2} \\
& \mathrm{~S}_{\mathrm{y}, \mathrm{i}}^{2}=\sum_{\mathrm{i}=1}^{\mathrm{n}}\left(\mathrm{y}_{\mathrm{i}}-\mathrm{c}_{5} \cdot \cos \left(\mathrm{c}_{0} \cdot \mathrm{t}+\mathrm{c}_{6}\right)-\mathrm{c}_{7} \cdot \mathrm{t}-\mathrm{c}_{8}\right)^{2}  \tag{3}\\
& \mathrm{~S}_{\mathrm{z}, \mathrm{i}}^{2}=\sum_{\mathrm{i}=1}^{\mathrm{n}}\left(\mathrm{x}_{\mathrm{i}}-\mathrm{c}_{9} \cdot \cos \left(\mathrm{c}_{0} \cdot \mathrm{t}+\mathrm{c}_{10}\right)-\mathrm{c}_{11} \cdot \mathrm{t}-\mathrm{c}_{12}\right)^{2}
\end{align*}
$$

where n is the number of observations (grid points in the supposed helix-like structure).

A simple look on (3) gives the minimal number of observations (n) to obtain a statistically significant result: $n \geq 3 \times 12 / 3$ (three coordinates - e.g. $x$, $y, z-12$ unknowns, 3 observations per unknown). Therefore, no less than 12 monomers should be in the polymer in order to obtain statistical significant coefficients.

For a double helix, one follows equations:

$$
\begin{align*}
& x=\left(c_{1}+(-1)^{t} \cdot c_{13}\right) \cdot \cos \left(c_{0} \cdot 2 \pi \cdot t+c_{2}+2 \pi \cdot(-1)^{t} \cdot c_{19}\right)+c_{3} \cdot t+c_{4}+(-1)^{t} \cdot c_{16} \\
& y=\left(c_{5}+(-1)^{\mathrm{t}} \cdot \mathrm{c}_{14}\right) \cdot \cos \left(\mathrm{c}_{0} \cdot 2 \pi \cdot \mathrm{t}+\mathrm{c}_{6}+2 \pi \cdot(-1)^{\mathrm{t}} \cdot \mathrm{c}_{19}\right)+\mathrm{c}_{7} \cdot \mathrm{t}+\mathrm{c}_{8}+(-1)^{\mathrm{t}} \cdot \mathrm{c}_{17}  \tag{4}\\
& z=\left(\mathrm{c}_{9}+(-1)^{\mathrm{t}} \cdot \mathrm{c}_{15}\right) \cdot \cos \left(\mathrm{c}_{0} \cdot 2 \pi \cdot \mathrm{t}+\mathrm{c}_{10}+2 \pi \cdot(-1)^{\mathrm{t}} \cdot \mathrm{c}_{19}\right)+\mathrm{c}_{11} \cdot \mathrm{t}+\mathrm{c}_{12}+(-1)^{\mathrm{t}} \cdot \mathrm{c}_{18}
\end{align*}
$$

As can be observed, for the double helix, a series of new unknown coefficients appeared; $c_{19}$ parameterizes a possible shift between the appearance of the helixes, while $\mathrm{c}_{16}, \mathrm{c}_{17}$ and $\mathrm{c}_{18}$ account for the differences among the origin points of the helixes, and finally $\mathrm{c}_{13}, \mathrm{c}_{14}$ and $\mathrm{c}_{15}$ parameterizes the differences in amplitude.

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